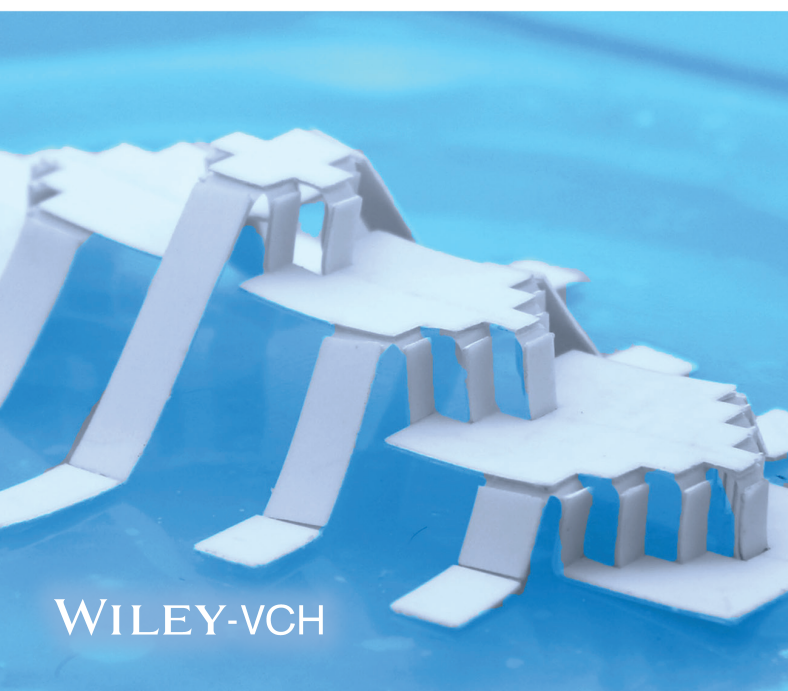
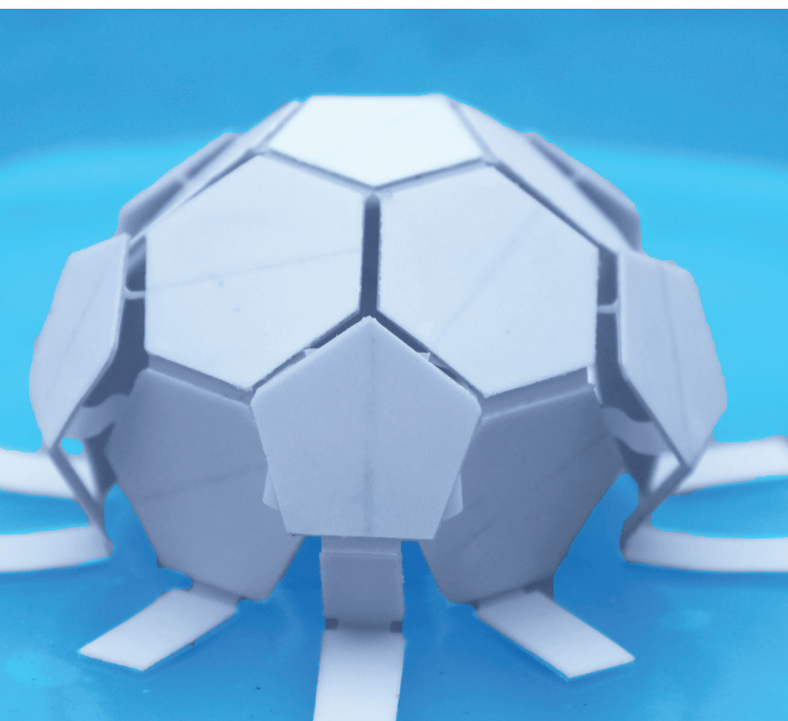


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# ADVANCED FUNCTIONAL MATERIALS



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# Controlled Mechanical Buckling for Origami-Inspired Construction of 3D Microstructures in Advanced Materials

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Origami is a topic of rapidly growing interest in both the scientific and engineering research communities due to its promising potential in a broad range of applications. Previous assembly approaches for origami structures at the micro/nanoscale are constrained by the applicable classes of materials, topologies, and/or capability for reversible control over the transformation process. Here, a strategy is introduced that exploits mechanical buckling for automatic origami assembly of 3D structures across material classes from soft polymers to brittle inorganic semiconductors, and length scales from nanometers to centimeters. This approach relies on a spatial variation of thickness in the initial 2D structures as a means to produce engineered folding creases during the compressive buckling process. The elastic nature of the assembly scheme enables active, deterministic control over intermediate states in the 2D to 3D transformation in a continuous and reversible manner. Demonstrations include a broad set of 3D structures formed through unidirectional, bidirectional, and even hierarchical folding, with examples ranging from half cylindrical columns and fish scales, to cubic boxes, pyramids, starfish, paper fans, skew tooth structures, and to amusing system-level examples of soccer balls, model houses, cars, and multifloor textured buildings.

## 1. Introduction

Origami, the ancient Japanese art of paper folding, involves the transformation of planar paper sheets into macroscopic 3D sculptures with diverse topologies.<sup>[1]</sup> Origami is now a topic of rapidly growing interest in the scientific and engineering research communities due to its potential for use in a broad range of applications, from self-folding microelectronics,<sup>[2–4]</sup> deformable batteries,<sup>[5,6]</sup> and reconfigurable metamaterials,<sup>[7–9]</sup> to artificial DNA constructs.<sup>[10,11]</sup> Important recent advances in the fundamental aspects of origami include the identification of mechanisms for bistability in deformed configurations,<sup>[7,12,13]</sup> and the development of lattice kirigami (a variant of origami that involves both cutting and folding) methods that solve the inverse problem of folding a flat plate into a complex targeted

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3D configuration.<sup>[14,15]</sup> In parallel, experimental methods are emerging for the assembly of origami structures at the micro/nanoscale. For example, a representative class of approaches relies on self-actuating materials, such as shape memory alloys,<sup>[16,17]</sup> shape memory polymers,<sup>[18–20]</sup> liquid crystal elastomers,<sup>[4]</sup> and hydrogels,<sup>[21–23]</sup> for programmable shape changes. These schemes are, however, not directly applicable to many technologically important types of materials, such as semiconductors or metals. Other routes rely on capillary forces (or surface tension forces),<sup>[24–27]</sup> or residual stresses in thin films<sup>[28–31]</sup> to drive the origami assembly, with the distinct practical advantage of compatibility with established planar device technologies. In most cases, however, such approaches<sup>[24,26–30]</sup> are irreversible and offer limited control of parameters such as the folding angle, or folding rate. The results described in the following introduce a set of concepts in which controlled mechanical buckling actuates origami type transformations, in a manner that is fully applicable across material classes from soft polymers to brittle inorganic semiconductors, and length scales from nanometers to centimeters. This assembly scheme also enables active, deterministic control over intermediate states in the 2D to 3D transformation, in a continuous and reversible manner. These ideas complement and extend those that exploit buckling of filamentary ribbons to yield 3D mesostructures with open mesh networks,<sup>[32,33]</sup> and of membranes with kirigami patterns of cuts<sup>[34]</sup> to form 3D mesostructures with closed curved surfaces, thereby opening the access to a wide range of 3D topologies with levels of geometrical complexity significantly beyond those reported previously. In particular, the previous designs<sup>[33,34]</sup> adopt ribbons/membranes with uniform thicknesses as the 2D precursors, thereby leading to deformations mainly in the form of global bending and twisting throughout the entire structures, as a result of the compressive buckling process. These layouts do not allow well-controlled folding deformations at desired local regions, as is essential to the notion of origami. The present study introduces a set of concepts that rely on a spatial variation of thickness in the initial 2D structures to produce engineered folding creases during the compressive buckling process, with capabilities in origami. Demonstrations include a broad set of 3D structures formed through unidirectional, bidirectional, and even hierarchical folding, with examples ranging from half cylindrical columns and fish scales, to cubic boxes, pyramids, starfish, paper fans, skew tooth structures, in addition to fanciful examples of soccer balls, model houses, cars, and multifloor textured buildings.

## 2. Results and Discussion

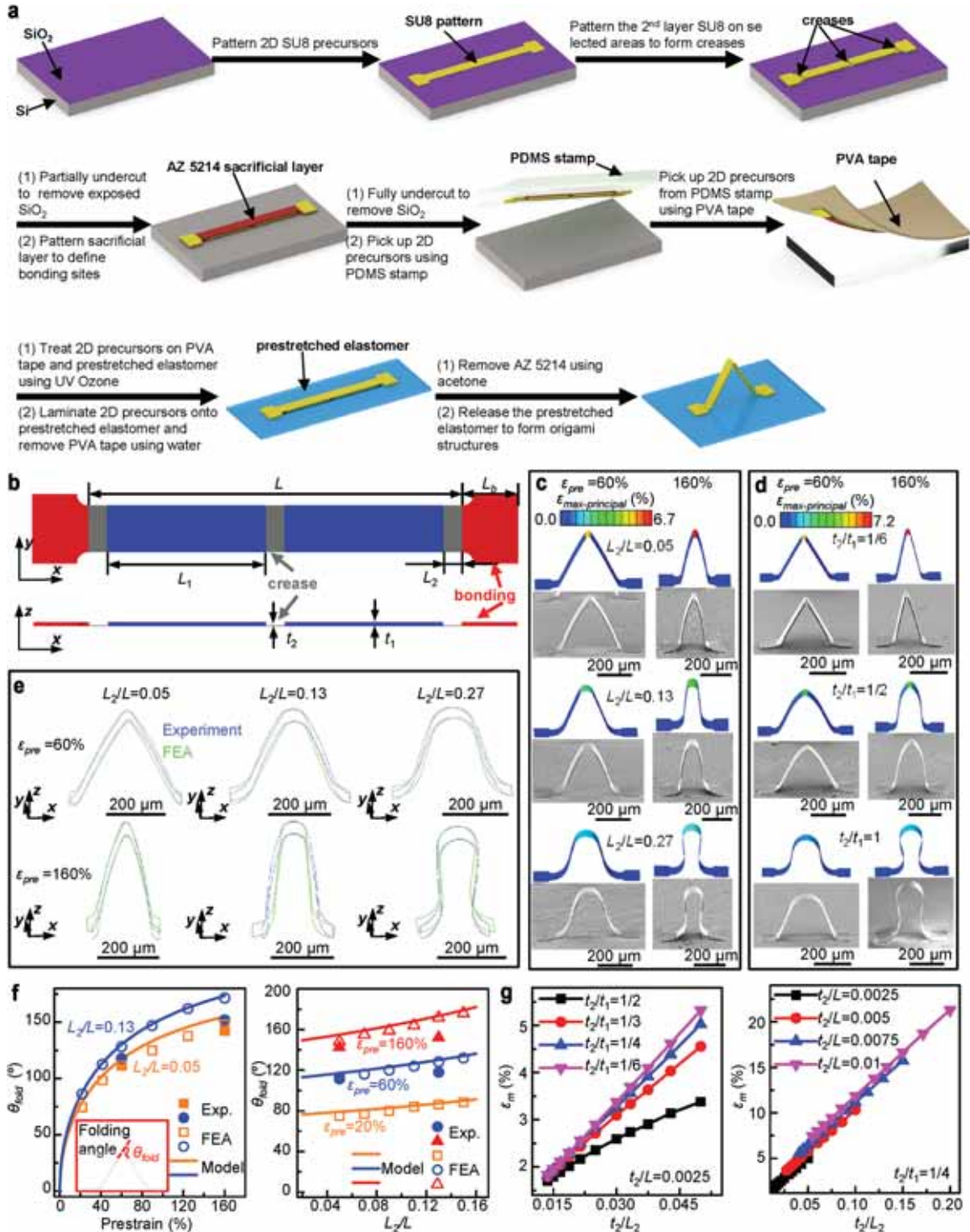
A straight ribbon serves as a simple example to illustrate the overall design concept (Figure 1a). Here, compressive forces associated with relaxation of prestrain in a supporting elastomer substrate exert on a ribbon structure (i.e., the 2D precursor in this case) at selected regions (i.e., bonding region, length  $L_b$ ; red, Figure 1b), thereby deforming the system into a 3D structure in an overall process similar to that used previously in ribbons/membranes with uniform thickness.<sup>[33,34]</sup> The key difference here is that the 2D precursor includes

engineered variations in thickness to guide folding deformations at specific, targeted locations. In this example, the ribbon (length  $L$ , excluding the bonding locations) consists of five segments (Figure 1b), two (in blue color, each with length  $L_1$  and thickness  $t_1$ ) of which are thicker than the other three (in gray color, each with length  $L_2$  and thickness  $t_2$ ). When the thickness ratio ( $t_2/t_1$ ) is relatively small (e.g.,  $<1/3$ ), the thick segments undergo negligible deformation while the thin ones accommodate the compression via folding, simply as a consequence of the cubic downscaling of the bending stiffness with thickness. Furthermore, as the length ratio ( $L_2/L$ ) decreases (e.g.,  $<0.1$ ), the radius of curvature associated with this folded region decreases. Figure 1c,d illustrates these behaviors through finite element analyses (FEA) and experimental observation for various length and thickness ratios. The FEA used the same dimensions as those in the corresponding experiments. Provided that material properties are independent of length scale, the calculated deformation and strain should remain the same if both the thickness and in-plane dimensions are scaled by the same factor. To provide a quantitative comparison, the borderlines of the scanning electron microscope (SEM) images were extracted, and overlaid with corresponding FEA results. Representative results appear in Figure 1e and Figure S1 (Supporting Information) for those configurations shown in Figure 1c,d. The data provide quantitative evidence of the good agreements between FEA and experiments. Some slight differences can appear mainly at regions very close to the bonding areas. Such discrepancies can be attributed to partial delamination. These results indicate that both a small thickness ratio ( $t_2/t_1 < 1/3$ ) and a small length ratio ( $L_2/L < 0.1$ ) are required to induce strong folding-type deformations in the thinner segments (referred to as creases in the following). Due to the large stiffnesses, the regions of substrate surface that are bonded with the 2D precursors undergo negligible deformations, such that the compressive strains are accommodated by the other nonbonded regions of substrate. For simplicity, the deformed configurations of creases can be assumed as arcs, while the thicker regions of the ribbons can be assumed to undergo negligible deformations, for relatively large thickness ratios (e.g.,  $t_2/t_1 > 1/4$ ). Then, the following relation can be obtained from geometric analyses

$$\frac{3L_2}{L} \frac{1}{\theta_{\text{fold}}} \sin \frac{\theta_{\text{fold}}}{2} + \left( \frac{1}{2} - \frac{3L_2}{2L} \right) \cos \frac{\theta_{\text{fold}}}{2} + \frac{L_b}{2L} - \frac{1}{2(1 + \epsilon_{\text{pre}})} \left( 1 + \frac{L_b}{L} \right) = 0 \quad (1)$$

which can be used to determine the folding angle ( $\theta_{\text{fold}}$ ) for a given prestrain. Figure 1f shows that the model predictions agree reasonably well with the FEA and experimental results. These results indicate that an enhanced level of prestrain increases the folding angle evidently.

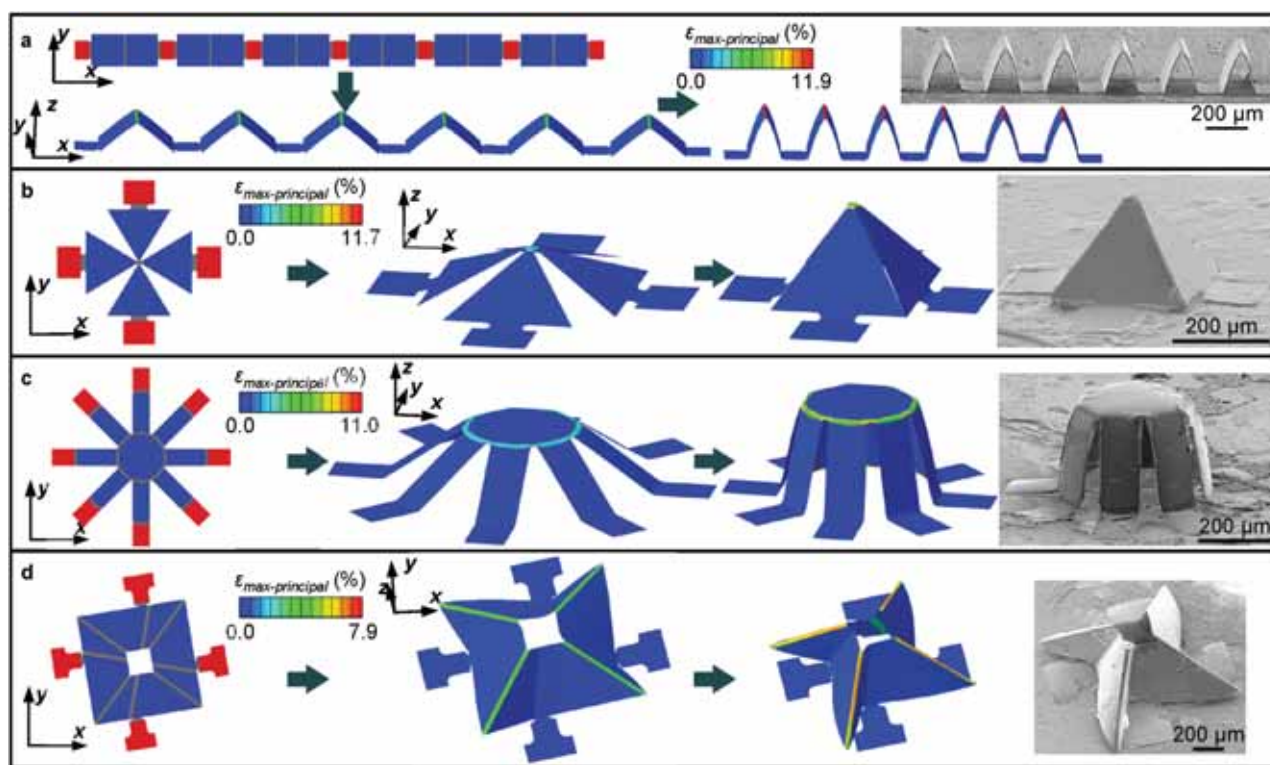
Since the overall compressive deformation is almost entirely accommodated by these creases, the maximum strains must also occur at these regions. Engineering designs that avoid fracture are important. For a straight ribbon with a small thickness ratio ( $t_2/t_1 \leq 1/4$ ), the peak value ( $\epsilon_m$ ) of the maximum principal strain ( $\epsilon_{\text{max-principal}}$ ) in the buckled 3D structures changes very slowly with further increase of  $t_2/t_1$  (Figure 1g, left), indicating that  $\epsilon_m$  becomes almost independent on  $t_2/t_1$



**Figure 1.** Buckling induced origami assembly of simple illustrative 3D structures. a) Schematic illustration of steps for fabricating 3D origami structures in polymer (SU8). b) Top and cross-sectional views of a straight ribbon with two different thicknesses. FEA results and corresponding SEM images of deformed mesostructures consisting of bilayers of metal (Au) and polymer (SU8) under two different levels (60% and 160%) of prestrain, for the nonuniform ribbons with a c) fixed thickness ratio ( $t_2/t_1 = 1/4$ ) and three different length ratios ( $L_2/L$ ), or with a d) fixed length ratio ( $L_2/L = 0.05$ ) and three different thickness ratios ( $t_2/t_1$ ). e) Borderlines of the deformed configurations for the mesostructures in (b), where the curves in green and blue correspond to results based on FEA and SEM images, respectively. f) Folding angle ( $\theta_{fold}$ ) versus the prestrain (left) for a straight ribbon with a fixed  $L_b/L = 0.15$  and two different length ratios, and versus the length ratio (right) for a straight ribbon with a fixed  $L_b/L = 0.15$  and three different prestrains. g) Maximum material strain versus the normalized thickness ( $t_2/L_2$ ) of a straight ribbon (b) under compression with 60% prestrain, for a fixed  $t_2/L = 0.0025$  with a range of  $t_2/t_1$  (left), and a fixed  $t_2/t_1 = 1/4$  with a range of  $t_2/L$  (right). In Figure c,d, the color in the FEA results corresponds to the magnitude of maximum principal strain. Scale bars: 200  $\mu$ m.

for sharp creases. Due to the nature of bending deformation in the creases, the maximum strain ( $\epsilon_m$ ) follows a proportional dependence on the normalized thickness ( $t_2/L_2$ ) for a given pre-strain ( $\epsilon_{pre}$ ), suggesting a general scaling law,  $\epsilon_m = F(\epsilon_{pre})t_2/L_2$ , where  $F(\epsilon_{pre})$  denotes a function that can be determined from FEA. This relation is well supported by FEA results in Figure 1g (right) for a fixed thickness ratio ( $t_2/t_1 = 1/4$ ) and a wide range of  $t_2/L$ . Reductions in the crease thickness ( $t_2$ ) and increases in the crease width ( $L_2$ ) act to reduce the maximum strains. Here, the prestrain ( $\epsilon_{pre}$ ) of the substrate in the case of uniaxial stretching can be defined as the average nominal strain ( $\epsilon_{11}$ ) of a uniform-strain area identified based on the criterion that the maximum difference of nominal strain  $\epsilon_{11}$  at this area is below  $\approx 10\%$  of the average value. Figure S2 (Supporting Information) shows that this prestrain is approximately equal to the applied strain under uniaxial stretching. The prestrain under biaxial stretching can be defined similarly, as shown in Figure S3 (Supporting Information), where we adopt  $\epsilon_{x-pre}$  and  $\epsilon_{y-pre}$  to denote the two components of normal strain along the two stretching directions. The distribution of maximum principal strain can involve some nonuniformities across the full area of the substrate, especially in the case of biaxial stretching. Assembly processes were typically performed in the central region of the substrate, where the strain distribution is uniform. Effects of nonuniform strains, to the extent that they are significant, can be incorporated explicitly into the FEA.

This basic strategy in thickness engineering affords great versatility in the design of mechanical buckling approaches to 3D structures. Figure 2 presents examples achieved with corresponding 2D precursors formed using a photodefinable epoxy (SU8; 2  $\mu\text{m}$  in thickness at the creases, and 6  $\mu\text{m}$  in thickness at the other region; Figure 2a,b), or with thin silicon (200 nm in thickness; Figures 2c,d) on top of the thick region. The left top frame of Figure 2a presents an example that includes six unit cells of large rectangles with creases (gray) located at the center and two ends. The periodically distributed small rectangles (red) adhere strongly to a uniaxially prestrained elastomeric substrate through covalent chemical bonding, while all other regions of the 2D precursors release from the substrate via elimination of a sacrificial interface layer (see the Experimental Section for details). Relaxing the prestrain provides compressive forces that trigger out-of-plane buckling to thereby induce folding deformations at the creases. The final configuration corresponds to that of triangular columns, as shown by the results (Figure 2a, the right two frames) of FEA and the SEM. Effects of fatigue do not appear in the origami structure after they are stretched to  $\approx 30\%$  strain repetitively at a frequency of  $\approx 1$  Hz for  $\approx 1000$  cycles (Figure S4, Supporting Information). The color of the FEA results denotes the magnitudes of maximum principal strains that localize at the creases. The peak values remain below the fracture thresholds of the epoxy. These analyses assume that the material properties of the creases are the same as those in the thicker regions. Therefore, the localized



**Figure 2.** Representative examples of origami assembly of 3D mesostructures from corresponding 2D ribbons/membranes. a) FEA results (left three panels) that describe the formation of triangular columns made of polymer (SU8), along with a corresponding SEM image (rightmost panel) of the final configuration. b–d) Similar results that describe the formation of a 3D pyramid (SU8), cylindrical shell (bilayer of silicon and SU8), and a complex “windmill” mesostructure (bilayer of silicon and SU8). The color in the FEA results corresponds to the magnitude of the maximum principal strain. Scale bars: 200  $\mu\text{m}$ .

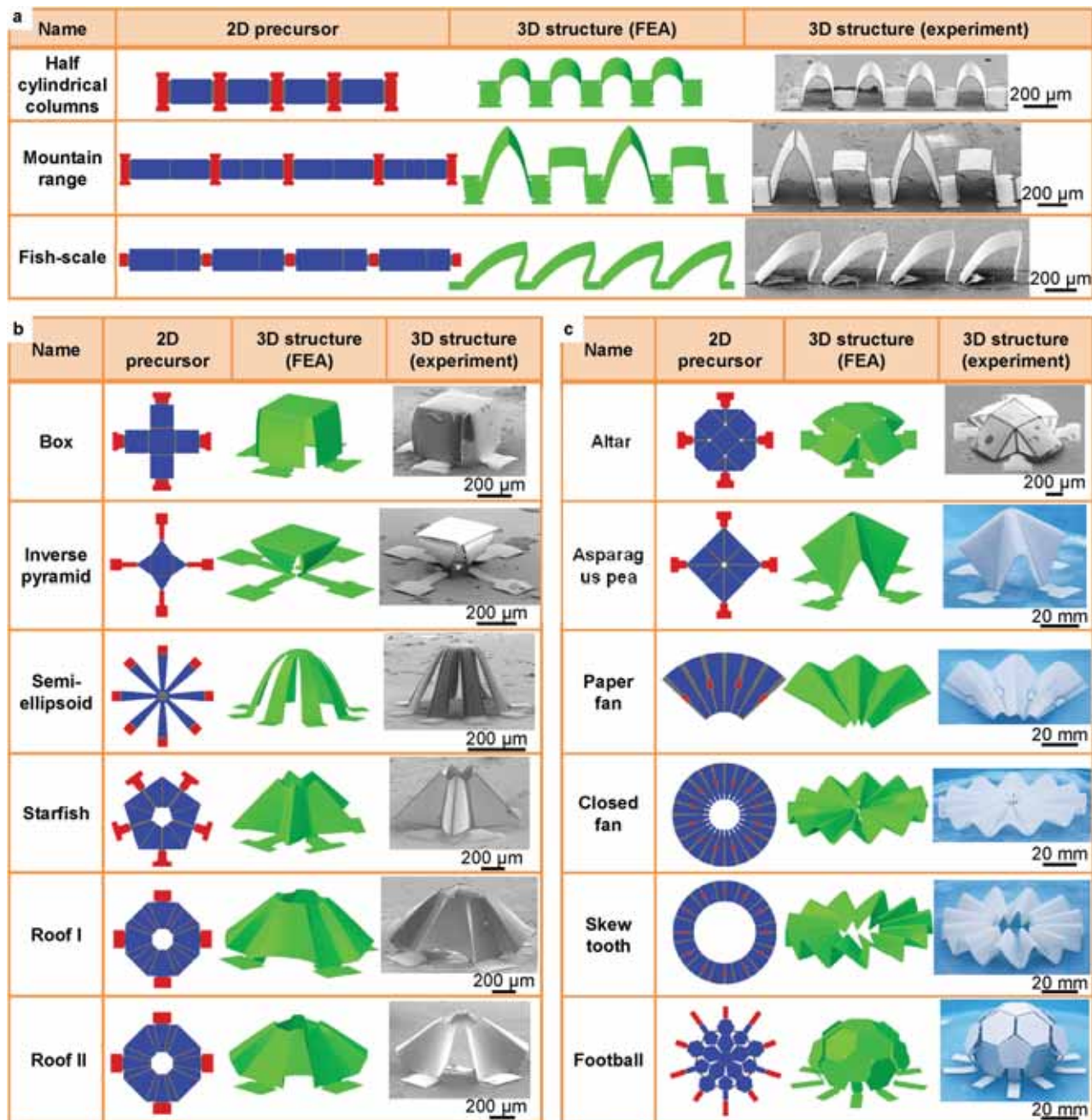
folding deformations result from the structural design with nonuniform thickness, and not from variations in materials properties. Delamination and fracture are the two main forms of failure during assembly. Such failures can be predicted, with reasonable accuracy, based on FEA. Here, our fracture criterion involves a strain threshold; FEA provides guidance on structure geometries that maintain the maximum principal strain below this threshold. Our delamination criterion relies on an energy based comparison of the undelaminated and delaminated states; again, FEA provides guidance. For the design parameters selected based on such calculations, we observe yields of  $\approx 50\%$  (i.e., half of the 2D precursors in a given fabrication run yield expected 3D structures) in the experiments. The failures arise from fracture and/or delamination associated with defects introduced in the fabrication process.

Figure 2b shows an example in which four triangular structures connect at the center with creases. Here, the assembly forms a pyramidal mesostructure, in which a carefully selected level of biaxial prestrain (in the substrate) yields a closed 3D topology. Many other configurations are possible with this scheme, including geometries that resemble a cylindrical shell (Figure 2c) and a windmill (Figure 2d). The latter case has a well-defined chirality, set by the configuration of creases. SEM images and corresponding FEA results of these 3D mesostructures viewed from different angles appear in Figure S5 (Supporting Information). The elastic tensile stiffnesses can be analyzed with FEA (Figure S6, Supporting Information), which indicates that the 3D pyramid offers a higher out-of-plane stiffness than those that resemble a cylindrical shell and a windmill. The physical nature of the assembly process allows its application across a broad range of material types and length scales. For example, all of the origami inspired mesostructures in Figure 2 can be reproduced using millimeter-scale plastic models, as shown by the optical images in Figure S7 (Supporting Information). The design guidelines associated with crease geometries gleaned from the analyses of straight ribbon geometries (Figure 1) also apply to cases of bidirectional folding, in that narrow regions with small thickness ratios ( $t_2/t_1 < 1/3$ ) lead to creases (Figure S8, Supporting Information). In all of the geometries examined herein, experimental results are in excellent agreement with FEA predictions, thereby establishing the computational approaches as reliable tools for rapid design optimization. Of note here are the stress concentrations that occur in the crease areas. For some 3D structures like structures in Figure 2, the maximum strain on the creases is even beyond 6%, higher than the fracture strain of a brittle material like silicon ( $\approx 2\%$ ). For device applications, a simple solution is to replace the silicon with stretchable serpentine interconnections in these regions.

A diverse range of basic origami inspired geometries, each identified with a descriptive name, can be realized with these approaches as summarized in Figure 3. Representative examples formed through uniaxial compression appear in Figure 3a, including geometries that resemble half cylindrical columns, fish scales, and columns with mixed cross-sectional shapes of triangles and rectangles. Origami mesostructures with more complex geometries can be realized using equal biaxial prestrain ( $\epsilon_{\text{pre}} = \epsilon_{x\text{-pre}} = \epsilon_{y\text{-pre}}$ ) in the substrate as demonstrated in Figure 3b,c, and Figure S9 (Supporting Information). These

shapes range from semiellipsoids and starfish, to polyhedral topologies (e.g., cube, inverse pyramid, football, etc.), to those (“closed fan” and “skew tooth”) formed through creases organized circumferentially in an annulus. In addition to the 2D layout of the precursor materials and the locations of the bonding, the patterns of creases provide additional means to control the final 3D geometries. With the same 2D precursor (e.g., an octagonal shape bonded at the centers of four edges), different distributions of creases yield distinct 3D configurations (e.g., ‘Roof I’ and ‘Roof II’ in Figure 3b). During the assembly process, the regions of substrate surface that are bonded with the 2D precursors undergo negligible deformations, such that the compressive strains are accommodated by the other nonbonded regions of substrate. As a result, for some of the origami structures (e.g., inverse pyramid and “Asparagus pea”) examined herein, large bonding areas are required to form the desired configurations with levels of biaxial prestrain that can be reproducibly achieved in the silicone materials used for the substrates, e.g.,  $\epsilon_{\text{pre}} < 200\%$ . With the inverse pyramid as an example, a moderate level ( $\approx 134\%$ ) of prestrain yields the structure in Figure 3b. With reduced dimensions (1/4 length of the original design) for the bonding areas, a substantially increased strain magnitude ( $\approx 320\%$ ) is necessary to yield the same 3D configuration (Figure S10, Supporting Information). All of the origami structures formed at microscale can also be constructed at millimeter (or larger) scales (Figure S11, Supporting Information) with plastic films.

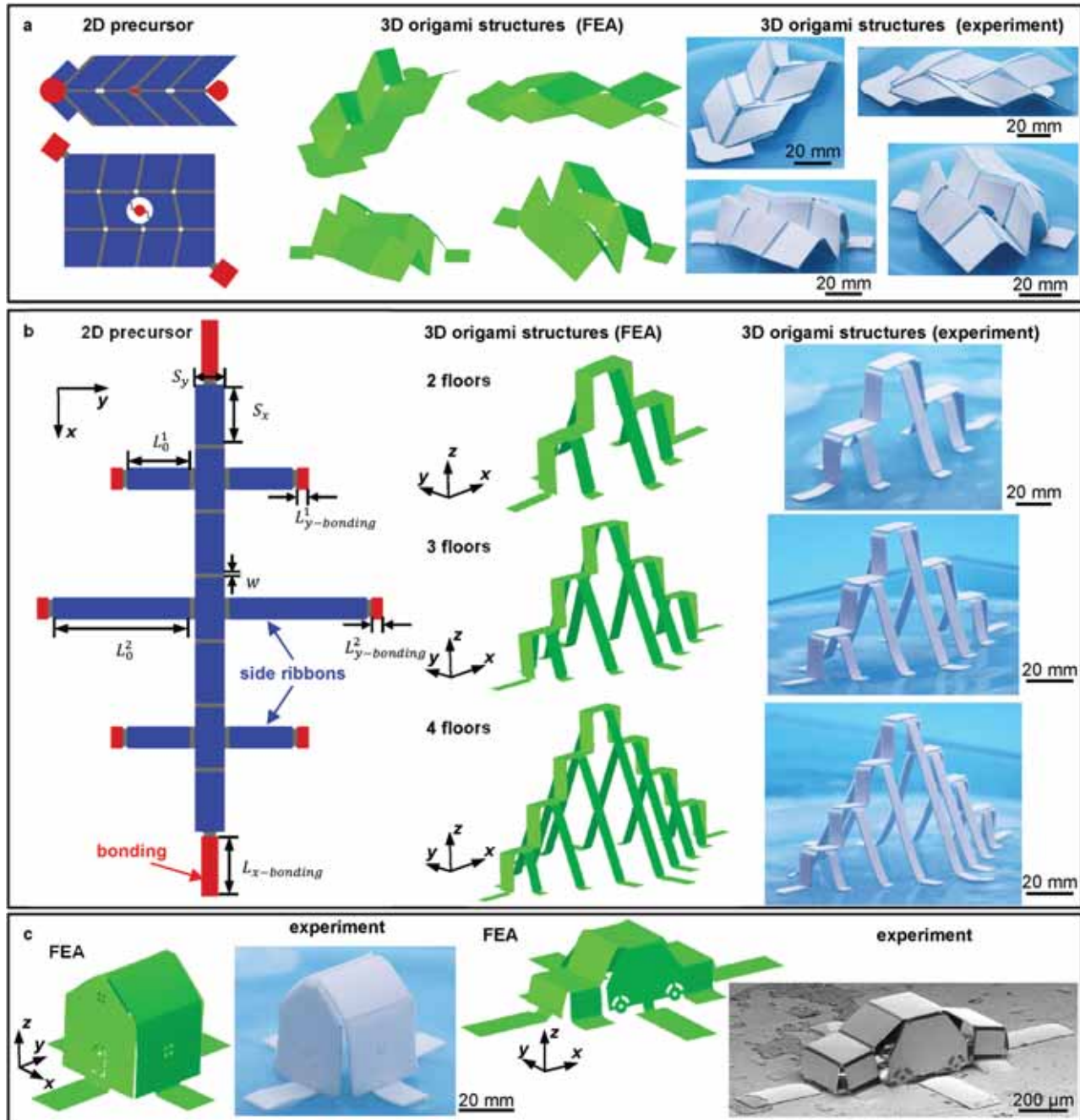
Multiple, hierarchical forms of folding are also possible with appropriate choices in design. Two examples of the well-known Miura-origami construct appear in Figure 4a, each of which uses three bonding locations to impart the compressive forces. In the first design (Figure 4a, top), the bonding locations lie at the intersection of differently oriented creases, such that creases originally along horizontal direction (in the 2D state) form a wavy configuration during assembly. In the second design (Figure 4a, bottom), the bonding locations distribute along the diagonal direction of the 2D precursor, such that the assembly tends to shrink the structure into a smaller (and thicker) structure. Examples of origami structures that have multilevel constructions in the out-of-plane direction appear in Figure 4b. Note that the folding of creases can occur in two different directions, based on energetic considerations, for cases of multifloor designs. This equivalence can be avoided by including supporting beams (e.g., the side ribbons in the left panel of Figure 4b) with optimized lengths to control the folding directions, thereby leading to consistent, desired multifloor structures. In this study, the strain energies of different possible modes were calculated, to reveal designs in which the strain energy of the targeted configuration is much lower than the other possible modes. Figure S12 (Supporting Information) shows that when a straight ribbon with eight creases serves as the 2D precursor, the two modes with multiple folds have similar levels of strain energies and that both are larger than the mode that takes the form of a single large arc. Practically, in the absence of defects that coincide with modes that have multiple folds, the single large arc is a unique solution. With the addition of supporting ribbons, a two-floor building in a stepwise configuration can be obtained, because the corresponding strain energy is much lower than the other possible



**Figure 3.** Experimental and computational studies of various origami structures. a) 2D precursors, FEA predictions and SEM images for three meso-structures formed with the use of uniaxial prestrain in the substrate. b,c) 2D precursors, FEA predictions, and experimental (SEM or optical) images for twelve structures formed with the use of biaxial prestrain in the substrate. Roof I in (b) is made of polymer (SU8); box and semiellipsoid in (b) are made of bilayers of silicon and polymer (SU8); the last five structures in (c) are made of thin plastic films; and all the other origami structures are made of bilayer of metal (Au) and polymer (SU8). The scale bars of SEM and optical images are 200  $\mu\text{m}$  and 20 mm, respectively.

modes, as shown in Figures S12c,d (Supporting Information). Figure 4b (left) illustrates the various length parameters in such a 2D precursor. Here, the  $x$ -directional ribbon consists of seven identical segments ( $S_x$  by  $S_y$ ) connected with creases ( $w$  in width). Adhesion to the elastomeric substrate occurs at two rectangular regions (red;  $L_{x\text{-bonding}}$  in the  $x$ -directional length). Three pairs of side ribbons with lengths of  $L_0^1$  or  $L_0^2$  connect the vertical ribbon with the bonding locations ( $L_{y\text{-bonding}}^1$

or  $L_{y\text{-bonding}}^2$  in the  $y$ -directional length), to assist in the formation of the multifloor design. To form a stepwise geometry in which each segment of the vertical ribbon orients perpendicular to the neighboring segments in the 3D shape, mechanics models (See the Supporting Information for details) that correlate the required prestrain values ( $\epsilon_{x\text{-pre}}$  and  $\epsilon_{y\text{-pre}}$ ) to various geometric parameters ( $S_x$ ,  $S_y$ ,  $w$ ,  $L_0^1$ ,  $L_0^2$ ,  $L_{y\text{-bonding}}^1$ ,  $L_{y\text{-bonding}}^2$ , and  $L_{x\text{-bonding}}$ ) and acute angles ( $\theta_1$  and  $\theta_2$ ) between the side ribbons



**Figure 4.** 3D structures formed through multiple, hierarchical forms of folding. a) 2D precursors, FEA predictions, and optical images for two structures involving Miura-origami. b) Schematic illustration of a representative 2D precursor for a two-floor building, FEA predictions, and optical images of stepwise structures with two to four floors. c) FEA predictions and experimental images for two origami structures (left frame, made of plastic; right frame, made of bilayer of metal (Au) and polymer (SU8)) that resemble a house and a car. The scale bars of SEM and optical images are 200  $\mu\text{m}$  and 20 mm, respectively.

and the  $y$ -axis in the final geometry are useful. Figure 4b (upper middle and upper right) provides an example of a two-floor building formed with a nonequal biaxial prestrain ( $\epsilon_{x\text{-pre}} = 96\%$ ,  $\epsilon_{y\text{-pre}} = 110\%$ ), which exhibits a clear stepwise configuration along the  $x$ -axis, with tilt angles of  $\theta_1 = 75^\circ$  and  $\theta_2 = 68^\circ$  for the side ribbons. Such designs can be, in principle, extended to origami structures with arbitrary numbers of floors (See the

Supporting Information for details). Two stepwise structures demonstrated in Figure 4b and Figure S13 (Supporting Information) have three and four floors, respectively, with tilt angles of  $\theta_1 = 75^\circ$  and  $\theta_2 = \theta_3 = \theta_4 = 68^\circ$ , for the side ribbons. For origami structures with multiple (e.g.,  $n > 3$ ) floors, theoretical analyses (Figures S14 and S15, Supporting Information) highlight challenges in realizing upright side ribbons (with  $\theta_i = 90^\circ$ ,



$i = 1, \dots, n$ ) experimentally, due to the practical difficulties in achieving, simultaneously, large bonding sizes ( $L_{y\text{-bonding}}^i$ ,  $i = 1 \dots n$ ) and large prestrains ( $\epsilon_{y\text{-pre}}$ ).

A further extension of these design concepts in multilevel origami allows the assembly of complex recognizable 3D objects. Figure 4c and Figure S16 (Supporting Information) illustrate structures that resemble a house and a car, respectively, achieved with tailored values of prestrain ( $\epsilon_{x\text{-pre}} = 119\%$  and  $\epsilon_{y\text{-pre}} = 138\%$  for the house;  $\epsilon_{x\text{-pre}} = 33\%$  and  $\epsilon_{y\text{-pre}} = 53\%$  for the car). The 2D precursors of these two designs appear in Figure S17 (Supporting Information). For complex origami geometries that involve hierarchical forms of folding, experimental results both in millimeter and microscales agree well with FEA predictions.

Concepts inspired by kirigami involve the addition of patterns of cuts to expand the accessible range of 3D structures. The concepts described previously are readily compatible with such approaches.<sup>[34]</sup> Figure 5a,b illustrates that cuts can play similar, but different, roles in the process compared to controlled variations in thickness. For example, patterns of cuts placed at appropriate locations, like those shown in Figure 5a (top), induce folding deformations at the corresponding locations to yield origami structures that are very close to the “starfish” in Figure 3b and “Asparagus pea” in Figure 3c. Such similar roles originate from the same effects of decreased bending stiffness at targeted regions through reduction of their width or thickness. Separately, the removal of material by introduction of cuts additionally reduces the effects of stress concentrations<sup>[34]</sup> and allows enhanced levels of flexibility in deformation (e.g., the twisting mode in Figure 5b), which are unobtainable through reductions in thickness (Figure S18, Supporting Information). Figure 5c–f and Figure S19 (Supporting Information) demonstrate that combinations of designs in creases and cuts enlarge substantially the accessible range of 3D topologies. The first structure of Figure 5c (left) corresponds to a closed 3D polyhedron that eliminates the diamond hole through contraction of surrounding surfaces. To its right is an origami structure that resembles a fence, formed mainly through rotation of two main components connected with short creases at the central region. Figure 5d presents two examples of lattice kirigami<sup>[14,15]</sup> implemented through the buckling guided approach. Here, bonding locations periodically distributed in a triangular lattice pattern transfer the compressive forces to the 2D precursors, deforming them into a hexagonal array of hexagonal prisms or triangular array of triangular prisms. More complex integration of crease and cut patterns appear in Figure S20 (Supporting Information), which forms a “Ziggurat” architecture (Figure 5e) by means of compressive buckling. Figure 5f corresponds to a variant of the three-floor building in Figure 4b, in which well-arranged cuts and creases form textured steps connecting the adjacent planes (parallel with the substrate surface). Such designs can extend to the assembly of configurations with a different number of floors, as illustrated in Figure S21 (Supporting Information).

The 3D structures formed with the above approach remain tethered to the substrate during the origami assembly, and involve no plastic deformation in designs selected using guidance from FEA modeling. As such, their 3D configurations can be dynamically and reversibly tuned by mechanically loading

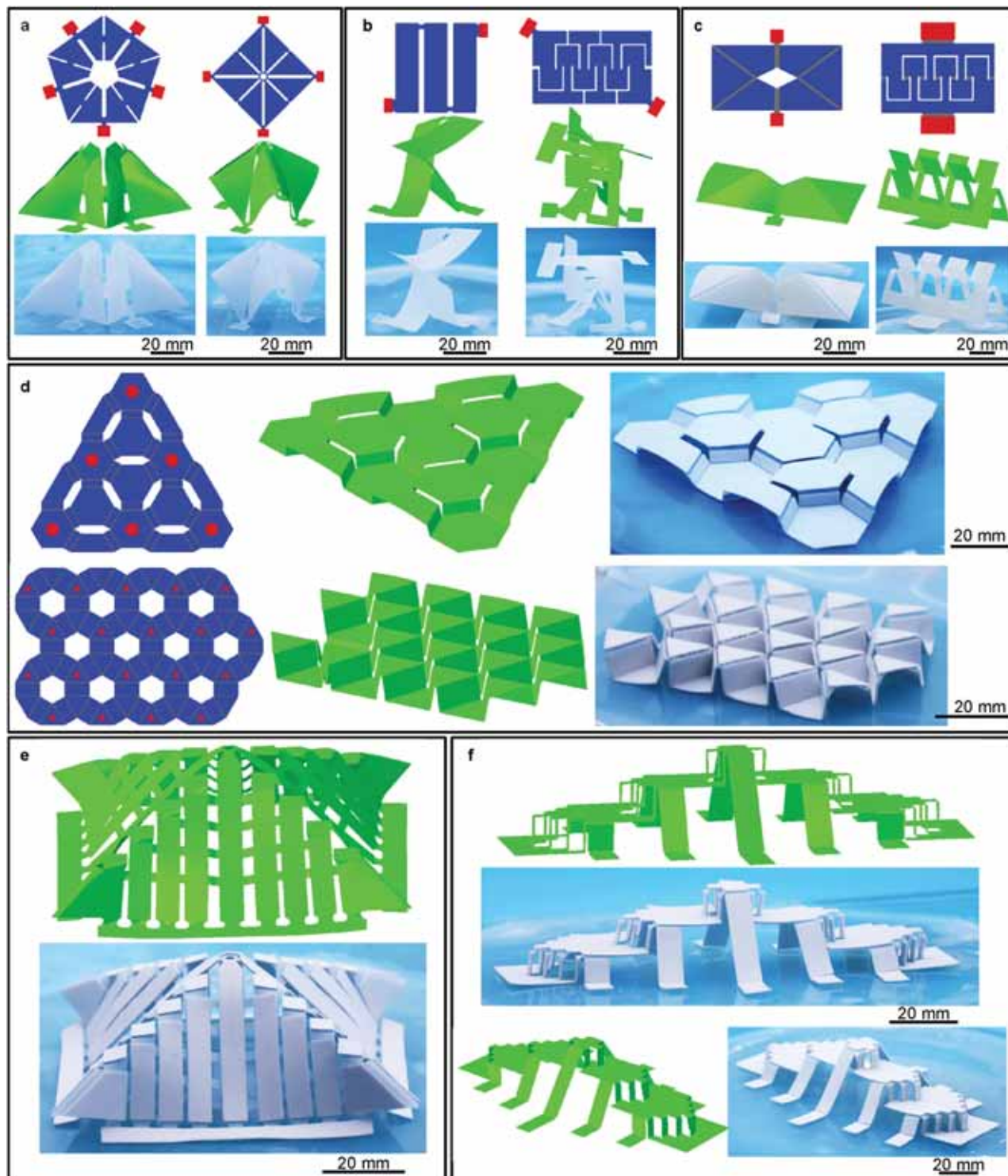
and unloading the substrate through stretching or bending. Figure 6 and Movies S1 and S2 (Supporting Information) provide two examples for the origami assembly of a soccer ball structure and a multifloor textured building structure that illustrate clearly such capabilities. Of particular note is the excellent agreement between the predictions from FEA modeling and the experiments for almost all of the intermediate states. Figure S22 (Supporting Information) illustrate quantitatively the continuous control of the configuration by using different levels of prestrain or applied strain in the substrate.

### 3. Conclusion

In summary, this paper introduces an approach that exploits controlled mechanical buckling for autonomic assembly of 3D structures across a wide range of length scales and material types, using a combination of origami inspired ideas, theoretical modeling methods and experimental techniques. These design concepts, combined with advanced microfabrication processes provide immediate access to 3D microarchitectures with a broad range of topologies in materials that include those that form the basis of advanced microsystems technologies in electronics, photonics, sensors, and microelectromechanical systems. These features allow transformation of virtually any type of existing 2D microsystem technology into a 3D configuration, thereby providing unusual design options in the development of fundamentally new kinds of devices. In addition, the assembly method allows continuous and reversible control of the 3D configurations over intermediate states in the 2D to 3D transformation. Such capabilities can be important in 3D electromagnetic components, such as inductors or antennas, where key properties (e.g., quality factors, resonant frequencies, directionality) can be adjusted in real time. These structures can also serve as the templates for the guided growth of materials or cells, or even as advanced types of single cell manipulators.

### 4. Experimental Section

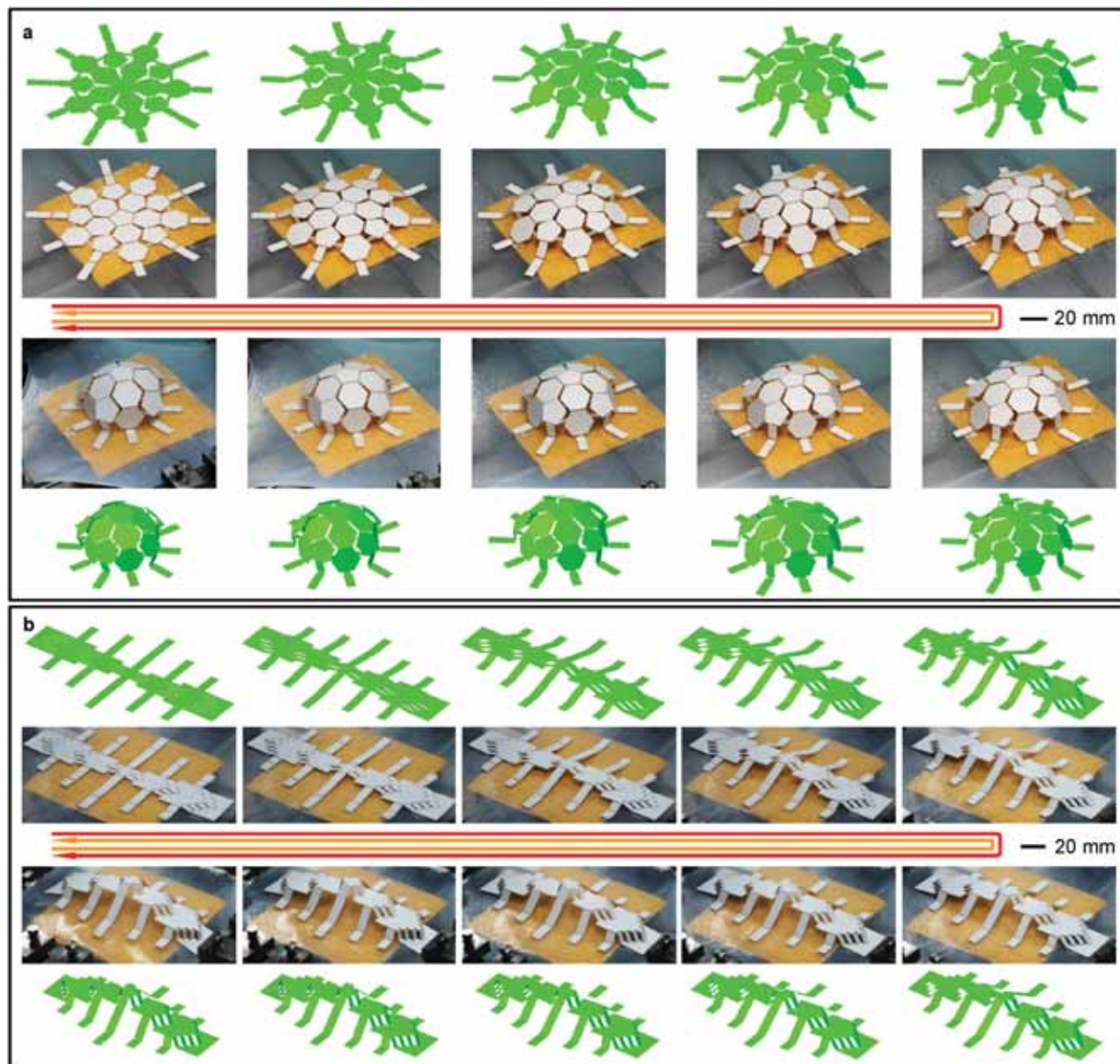
Preparation of origami mesostructures in polymer began with thermal oxidation to form a layer of silicon dioxide ( $\text{SiO}_2$ , 500 nm in thickness) on a silicon wafer. Next, spin casting and photolithography formed patterns of photodefinable epoxy (SU8, 2  $\mu\text{m}$  in thickness) on the  $\text{SiO}_2$ . Spin casting and photolithography with a second layer of SU8 (4  $\mu\text{m}$  in thickness) yielded 2D precursors with strategically designed variations in thickness (i.e., 2 or 6  $\mu\text{m}$ ). Immersion in hydrofluoric acid (HF) removed the buried  $\text{SiO}_2$  layer from the exposed regions and also slightly from under the edges of the SU8. Spin casting and photolithography created patterns of photoresist (AZ 5214, 1.6  $\mu\text{m}$  in thickness) on top of the SU8 layers to define the bonding sites. Immersion in HF eliminated the remaining  $\text{SiO}_2$  by complete undercut etching. The techniques of transfer printing enabled retrieval of the 2D precursors and their delivery to a piece of water soluble tape (polyvinyl alcohol, PVA). A thin sheet ( $\approx 0.5$  mm) of silicone elastomer (Dragon Skin, Smooth-On) served as the assembly platform, stretched to well-defined levels of prestrain (either uniaxial or biaxial) using a customized stage (Figure S23a, Supporting Information). Exposing the prestrained elastomer and the 2D precursors (on PVA) to UV-induced ozone yielded hydroxyl termination on their exposed surfaces. Laminating the tape onto the elastomer substrate with the exposed SU8 side down, followed by baking in an oven at 70  $^\circ\text{C}$  for 10 min yielded strong



**Figure 5.** 3D structures inspired by origami and kirigami concepts, through the uses of creases and/or cuts. a) 2D precursors, FEA predictions, and optical images for two folding structures, transformed from kirigami designs with uniform thicknesses. b) Similar results for two structures that involve not only folding but also twisting, also transformed from kirigami designs with uniform thicknesses. c,d) Similar results for four origami structures, transformed from designs with combined uses of cuts and creases (through nonuniform thicknesses). e,f) Two highly complex 3D structures: a “Ziggurat” architecture, and a three-floor building with textured steps. Scale bars: 20 mm.

covalent bonds via condensation reactions of surface-enriched hydroxyl groups between the elastomer substrate and the exposed regions of 2D precursors. After washing with hot water, acetone dissolved the PVA

tape and the photoresist. Slowly releasing the prestrain completed the origami assembly process. A schematic illustration of steps appears in Figure S24 (Supporting Information).



**Figure 6.** Tunable and reversible control of 3D origami type structures based on the strain applied to the elastomer assembly substrate. Optical images and FEA predictions that illustrate the reversible process of 2D–3D origami assembly guided by compressive bulking, for a) a plastic soccer ball and b) a plastic multifloor textured building structure.

Preparation of mesostructures of bilayers of silicon and polymer (SU8) began with photolithographic patterning and reactive ion etching of the top silicon layer (200 nm in thickness) to form silicon patterns on a silicon-on-insulator wafer. Next, spin casting and photolithography defined the geometries of the first layer of SU8 (2  $\mu\text{m}$  in thickness) on the silicon patterns. Spin casting and photolithography of a second layer of SU8 (4  $\mu\text{m}$  in thickness) in a matching geometry with silicon patterns yielded 2D origami precursors of silicon and SU8 bilayers. The remaining steps of photoresist patterning, HF undercutting, transfer printing, and 3D assembly followed procedures described above. A schematic illustration of steps appears in Figure S25 (Supporting Information).

Preparation of mesostructures of bilayers of metal (gold) and polymer (SU8) began with thermal oxidation to form a layer of  $\text{SiO}_2$  (500 nm in thickness) on a silicon wafer. Next, photolithography, electron beam evaporation deposition, and liftoff defined patterns

of gold (50 nm in thickness) as 2D precursors on the  $\text{SiO}_2$ . Spin casting and photolithography defined the geometries of the first layer of SU8 (2  $\mu\text{m}$  in thickness) on top of the gold. Spin casting and photolithography of a second layer of SU8 (4  $\mu\text{m}$  in thickness) yielded 2D origami precursors of gold and SU8 bilayers. The remaining steps of photoresist patterning, HF undercutting, transfer printing, and 3D assembly followed procedures described above. A schematic illustration of steps appears in Figure S26 (Supporting Information).

Preparation of structures in plastic films began with mechanical cutting of a base layer ( $\approx 75 \mu\text{m}$  in thickness) into desired patterns, followed by cutting of patterns to define the thick regions with additional layers of this same material. Most of the plastic structures used a thickness ratio of 1/4, except for the football (Figure 3c), Miura-origami structures (Figure 4a), multifloor buildings (Figures 4b and 5f), lattice kirigami structures (Figure 5d), and “Ziggurat” architecture (Figure 5e),

which used a thickness ratio of 1/8. Adhering additional layers of plastic films onto the base layer through thin, stick double-coated tape (9080A, 3M, Minnesota, USA) yielded 2D precursors for assembly using a prestrained silicone substrate (Figure S23b, Supporting Information). A highly viscous, organic adhesive, i.e., an RTV silicone rubber (706, NAN DA, Jiangsu, China), dispensed at desired locations on the 2D precursors resulted in strong bonding to the silicone substrate, after curing for  $\approx 2$  h at room temperature. Slowly releasing the prestrain in the substrate completed the assembly process.

All experiments involved a low rate of prestrain release, e.g.,  $>85$  s (for millimeter-scale experiments) and  $>120$  s (for millimeter-scale experiments) for full release of  $\approx 100\%$  prestrain, corresponding to a strain rate  $<0.012$  s $^{-1}$  (for millimeter-scale experiments), and  $<0.0084$  s $^{-1}$  (for microscale experiments). Under such circumstances, viscoelastic effects can be neglected, as confirmed by quantitative agreement between experiment and finite element computation that adopts hyperelastic constitutive law for the substrate.

## Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

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